Synthesis and study of properties of Zn nanotubes

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¹Institute of Nuclear Physics, Almaty, Kazakhstan ²L.N. Gumilyov Eurasian National University, Astana, Kazakhstan *E-mail: artem88sddt@mail.ru This paper describes the method of synthesis of Zn-based nanotubes using electrochemical deposition, as well as characterizes their morphological and conductive properties. Track membranes based on polyethylene terephthalate (PET) with a thickness of 12 microns with pore diameters of 380±10 nm were used as templates. Characterization of the structural features was carried out by scanning electron microscopy (SEM), energy dispersive analysis (EDA) and X-ray diffractometric analysis (XRD), and indirectly, during the study of electrical conductivity.

Calculation of the average crystallite sizes showed that the reduction in the intensity of the oxide phase ZnO reduces the size of the crystallites from 41.52 nm at 1.25 V to 29.34 nm at 1.75 V. The number of defects, which interferes with the movement of electrons, reduced with the decreasing of the average size of crystallites in the nanotubes. The number of defects directly affected the conductive properties of Zn based nanotubes.

Keywords: track-etched membranes; electrochemical deposition; nanotubes.

Zn-нанотүтікшелерін алу және қасиеттерін зерттеу

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¹Қазақстан Республикасының Ядролық физика институты, Алматы қ., Қазақстан ²Л.Н. Гумилев атындағы Еуразия ұлттық университеті, Астана қ., Қазақстан *E-mail: artem88sddt@mail.ru Мақалада электрохимиялық тұндыру арқылы Zn негізделген нанотүтікшелердің синтездеу әдісі сипатталды, сондай-ақ, олардың морфологиялық және өткізгіш қасиеттеріне мінездеме берілді. Үлгі ретінде қуыстықтардың диаметрі 380±10 нм, қалыңдығы 12 мкм полиэтилентерефталат (ПЭТ) негізінде трек мембраналары пайдаланады. Құрылымдық ерекшеліктерін сипаттау растрлық электрондық микроскопия (РЭМ), энергия-дисперсиялық талдау (ЭДТ) және рентгенодифрактометриялық талдау (РДТ) әдістермен өткізгішеді, және жанама электр өткізгіштігінің зерттеу барысында. Алынған нанотүтікшелердегі кристалиттердің орташа мөлшерін есептеу нәтижесінде ZnO оксидтік фазасының қарқындылығының төмендеуі кристалиттердің мөлшері 1,25 В-тағы 41,52-ден 1,75 В-тағы 29,34 нм дейін төмендейді. Алынған нанотүтікшелердегі кристаллитердің орташа мөлшерін азайту электрондардың қозғалысына кедергі болатын ақаулар санын азайтады. Ақаулар саны Zn нанотүтікшелерінің өткізгіш қасиеттеріне тікелей әсер етеді.

Түйін сөздер: тректік мембрана; электрохимиялық тұндыру; нанотүтіктер.

Получение и исследование свойств Zn – нанотрубок

^{1,2} Козловский А.Л.*, ²Мейримова Т.Ю., ^{1,2}Шлимас Д.И., ^{1,2}Здоровец М.В., ²Кадыржанов К.К.

¹Институт ядерной физики Республики Казахстан, г. Алматы, Казахстан ²Евразийский национальный университет имени Л.Н. Гумилева, г. Астана, Казахстан *E-mail: artem88sddt@mail.ru В данной работе описывается метод синтеза нанотрубок на основе Zn при помощи электрохимического осаждения, а также проводится характеризация их морфологических и проводящих свойств. В качестве шаблонов используются трековые мембраны на основе полиэтилентерефталата (ПЭТФ) толщиной 12 мкм с диаметрами пор 380±10 нм. Характеризация структурных особенностей проводится методами растровой электронной микроскопии (РЭМ), энергодисперсионного анализа (ЭДА) и рентгенодифрактометрического анализа (РСА), и, косвенно, при изучении электропроводности. Расчеты среднего размера кристаллитов показали, что снижение интенсивности пиков оксидной фазы ZnO на рентгенограмме приводит к уменьшению размеров кристаллитов с 41,52 нм для 1,25 В до 29,34 нм для 1,75 В. С уменьшением среднего размера кристаллитов в полученных нанотрубках снижается количество дефектов, которые препятствуют движению электронов. Количество дефектов напрямую влияет на проводящие свойства Zn – нанотрубок.

Ключевые слова: трековые мембраны; электрохимическое осаждение; нанотрубки.



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1. Introduction

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At present time, significant efforts are focused on synthesis and study of properties of metal one-dimensional nanomaterials or metal nanotubes/nanowires such as Au, Ag, Cu, Fe, Co, Ni, Pd, Bi, W [1-7]. This tendency is due to prospect of their application in optoelectronics, photo electrochemistry, catalysis, medicine etc. [8, 9]. The ability to monitor and control physical and chemical properties of such structures attract interest to them, because this property contributes to the expansion of use of nanomaterials.

There are only few works devoted to synthesis of zinc-based nanostructures along with other elements, despite a number of attractive features these structures have. For example, Zn is biologically safe, it has low toxicity and biocompatibility, therefore it can be applied in various configurations of nanostructures (nanowires, nanotubes, nanoparticles) [10, 11]. Zn-based nanomaterials have semiconducting and piezoelectric properties. This phenomenon is the basis for electrochemical related sensors and transducers. Practical application of semiconductors based on zinc or zinc oxide is only limited by a lack of methods that provide control over production of nanotubes with crystal structure of walls. Zink-based nanowires were obtained in the work [12] and nanoparticles – in the work [13].

There is a variety of methods used for preparation of nanostructures based on Zn and ZnO: chemical vapor deposition of metals [3], method of arc discharge [4], laser evaporation [5] and templated synthesis [6, 7]. The most acceptable method among them is method of templated synthesis due to satisfying requirements of stability and chemical inertness. The method of templated synthesis allows obtaining round pores with the same diameter from tens to hundreds of nanometers in template matrices, which are convenient materials for production of structurally homogeneous nano replicas.

In this paper, we describe the method of synthesis of

Zn-based nanotubes using electrochemical deposition and characterize their morphological and conductive properties. Track-etched membranes based on polyethylene terephthalate (PET) with thickness of 12 microns and pore diameters 380 ±10 nm were used as templates. Characterization of the structural features is carried out by scanning electron microscopy (SEM), energy dispersive analysis (EDA) and X-ray diffractometric analysis (XRD), and indirectly, during the study of electrical conductivity.

Successful production of Zn nanotubes will give possibility of their full-scale application as semiconductor materials in electronic and optoelectronic circuits [1, 2, 14, 15,] due to large width of bandgap (3.37 eV), piezoelectric properties and high energy of exciton bond (60 MeV).

2. Experiment

Track-etched membranes based on PET polymer matrix Hostaphan[®] manufactured by Mitsubishi Polyester Film (Germany) were used for producing Zn nanotubes by electrochemical deposition from electrolyte solution. PET film was irradiated by krypton ions using cyclotron DC-60 with the energy of 1.75 MeV/nucleon. UV-sensitization of irradiated track membranes was performed using UV-C lamp with a wavelength of 253.7 nm from each side for 30 min. UV radiation led to photo oxidation of surface layer of polymer and allowed obtaining pores with the same diameter in a polymer. After UV sensitization, film was chemically etched from both sides in 2.2 M NaOH aqueous solution at the temperature of 85±1°C for 4.5 min, followed by neutralization in 1.0% aqueous solution of acetic acid [16, 17].

Electrochemical deposition in tracks of template matrix was carried out at voltage of 1.25-1.75 V with a step of 0.25 V in the potentiostatic mode. Monitoring the progress of the experiment was done by chronoamperometry on multimeter Agilent 34410A.

Investigation of a structure and dimensions of obtained nanotubes was carried out using scanning electron microscope (SEM) Hitachi TM3030 (Japan) with microanalysis system Bruker XFlash MIN SVE (Germany) at accelerating voltage of 15 kV.

X-ray diffractometer studies were performed on the diffractometer Bruker D8 ADVANCE (Germany) equipped with X-ray tube with Cu-anode and graphite monochromator on the diffracted beam. Tube operation mode was 40 kV at 40 mA. Diffractograms were recorded in the angle range $20 - 120^{\circ} 2\theta$, step 0.03° and accumulation time 9 s.

Current source HP 66312A and ammeter Agilent34401A were used to measure the volt-ampere characteristics.

Before studying structural features of samples, the template matrix was removed by dissolving it in a hot solution of 9.0 M NaOH and then treated for 10 s in an ultrasonic bath to separate from polymer residues.

3. Results and discussions

PET-based matrix with pore density of 4.0 E+07 pores/cm² and thickness 12 microns was used as a template. The important feature of a templated synthesis is a possibility to control and regulate properties of obtained nanostructures due to using pores with a given geometry [6, 7, 17]. Diameter of pores of track membrane used for deposition of Zn nanotubes was 380±10 nm. Figure 1 shows images of track-etched membranes after 4.5 minutes of etching with pore diameters 380±10 nm obtained using scanning electron microscope.

Unilateral magnetron sputtering of a gold layer (thickness of 10 nm) on template matrices (Figure 1d) was done before electrochemical deposition of metal. Conductive layer was formed on the polymer as a result of magnetron sputtering in vacuum, which served as working electrode, a cathode, during deposition. Thickness of 10 nm allows formation of a substrate that does not cover pores entirely. It leads to the initial growth of structures along the walls during deposition process, ultimately forming hollow structures.

The electrolyte composition for producing Zn nanotubes is following: $ZnSO_4$, $7H_2O - 360$ g/l; $NH_4CI - 30$ g/l; $3H_2O$, $CH_3COONa - 15$ g/L; ascorbic acid – 120 g/L. Deposition of metals in pores of template matrix was carried out at potential difference from 1.25 to 1.75 V with step 0.25 V. Figure 2a shows dependence of amperage on the time of deposition of nanostructures during electrochemical deposition process.

It is known that the process of electrochemical deposition of metals in pores consists of 3 basic stages The first stage corresponds to the start of template pores filling that happens at t<50 s in our case. This stage is characterized by a sharp drop of current, which occurs as a result of nucleation process on cathode. The straight line on figure 2a corresponds to the second stage; at this stage, growth of nanotubes within pores happens. Process continues until the moment, when length of the nanotube becomes equal to the thickness of template. At the third stage, after reaching surface of template, metal starts growing in three dimensions (volume growth) above surface of polymer film, forming "caps" that grow from walls of nanotubes. As a result, effective surface area of deposition increases, as does the registered current. The third stage continues until overlap of metal islands located in areas of localization of nanotubes. Finally, continuous metal layer forms on the surface of template leading to the saturation of *I*. This value of *I* corresponds to the current of synthesis measured on metal electrode of the same



Figure 1 – SEM-images of PET-film after 4.5 min of etching. The surface of track membrane (a, b), a side view of the section of track membrane (c) and scheme of gold sputtering (d)



Figure 2 - Chronoamperogramms of deposition process (a) and scheme of nanotubes growth (b)

area and at the same applied voltage. It should be noted, that behavior of I(t) on chronoamperograms depends on parameters of deposition and shape of pores.

Calculations of volume rate of deposition were made using obtained chronoamperogramms. Determination of rate constant of deposition for potential difference allows controlling the time of template matrix filling and the height of nanostructures (Figure 3)



Figure 3 – Dependence of volume change rate on potential difference

The following conclusion can be made from calculations: volume rate of Zn deposition in pores of template matrix with diameter 380±10 nm increases exponentially with growth of potential difference from 1.25 V to 1.75 V.

The outer diameter of nanotubes was set by diameter of matrix pores and was 380±10 nm; the inner diameter was determined by method of gas permeability [18] to be 180±10 nm and the thickness of walls of nanotubes was 100±5 nm.

Figures 4a, 4c and 4e show images of Zn nanotubes. The height of nanotubes was determined from analysis of SEM-

images to be 12 microns; diameters of nanotubes correspond to diameter of pores of template matrices. Figures 4b, 4d and 4f represent EDA spectra of Zn nanotubes deposited at different potential differences.

Table 1 shows data on atomic ratio in obtained samples.

Table	1 -	Data	on	the	atomic	ratio	of	metals	in	obtained
nanostructures										

Name of sample	Atomic ratio
Zn 1.25 V	Zn ₇₃ O ₂₇
Zn 1.50 V	Zn ₈₆ O ₁₄
Zn 1.75 V	Zn ₁₀₀

These data indicate the formation of oxide phase of zinc with atomic ratio $Zn_{73}O_{27}$ occurring as a result of deposition at the potential difference of 1.25 V. Oxide phase is displaced with increasing applied potential difference – it happens due to a predominance of half of redox potential of Zn recovery (1.6 V) on formation of oxide phase ZnO (1.0 V). Thus, pure zinc nanotubes without additives of oxide phases can be obtained by electrochemical deposition at potential difference of 1.75 V.

X-ray diffractometer studies were carried out to identify crystal structure of Zn nanotubes. Diffractograms were recorded in the angle range $20 - 120^{\circ} 2\theta$ with step 0.03° and 9 s accumulation at each point. XRD experiment showed that samples have hexagonal primitive (HP) structure with lattice parameters different from reference values. Figure 5 shows X-ray diffractograms of test samples at 1.25, 1.5 and 1.75 V.

Presence of oxide phase ZnO is observed on diffractogram at 37°, 58°, 68°. A decrease in peak intensity of oxide phase ZnO is observed with increase of applied potential difference in crystal structure; oxide phase ZnO is not observed on diffractogram at potential difference 1.75 V, which is also confirmed by



Figure 4 – SEM images of Zn nanotubes (a, c, e); EDA spectra of Zn nanotubes at 1.25 V (b), 1.5 V (d), and at 1.75 V (f); SEM image of inner diameter of tubes (g)



data from energy dispersive analysis. Table 2 shows X-ray data of tested samples, parameters a and c of lattice and results of calculation of average sizes of crystallites by Scherrer equation (1):

$$\tau = \frac{k \lambda}{\beta \cos \theta},$$
 (1)

where k = 0.9 is a dimensionless coefficient of particle shape (Scherrer constant), $\lambda = 1.54$ Å is X-rays wavelength, β is a half-width at half-maximum (FWHM); ϑ is an angle of diffraction (Bragg angle). Size of crystallites was calculated for peaks (002), (100), (101), (112), (200), (201), (203). After that, an average value was calculated (Table 2).

Calculations of average sizes of crystallites showed that decrease of intensity of ZnO oxide phase leads to decrease of sizes of crystallites from 41.52 nm for 1.25 V to 29.34 nm for 1.75 V.

Texture coefficients were calculated by Harris formula (2):

$$TC(hkl) = \frac{I(hkl)}{I_0(hkl)} / \frac{1}{n} \sum \frac{I(hkl)}{I_0(hkl)}, \qquad (2)$$

where I(hkl) - relative intensity obtained experimentally; $I_o(hkl)$ - relative intensity corresponding to given plane according to base JCPDS; n – the number of planes. Table 3 shows results of calculations.

Calculations of texture coefficients in nanotubes synthesized at potential difference of 1.75 V show that texture of polycrystals becomes multicomponent (101), (112), (201).

Table 2 - Data from calculations of average sizes of crystallites

Name of sample	Parameters of cell sample	Data by base (No. of card)	Average size of crystallites, nm
Zn 1.25 V	Phase Zn (hexagonal primitive cell): a = (2.6650±0.0008) Å, c = (4.9273±0.0050) Å	No. 04-0831: a = 2.665 Å, c = 4.947 Å	41.52±1.3
Zn 1.50 V	Phase Zn (hexagonal primitive cell): a = (2.6648±0.0009) Å, c = (4.9289±0.0053) Å	No. 04-0831: a = 2.665 Å, c = 4.947 Å	37.15±1.5
Zn 1.75 V	Phase Zn (hexagonal primitive cell): a = (2.6655±0.0006) Å, c = (4.9231±0.0045) Å	No. 04-0831: a = 2.665 Å, c = 4.947 Å	29.34±1.2

Table 3 - Data of textural coefficients

209	(64)	TC(hkl)				
20=	(hkl)	Zn ₇₃ O ₂₇	Zn ₈₆ O ₁₄	Zn ₁₀₀		
36.296	002	0.840061	0.724619	0.703054		
38.992	100	1.045106	0.475531	1.047087		
43.231	101	3.408915	1.264461	3.278464		
82.102	112	0.535298	0.363604	1.083193		
83.765	200	0.356127	-	0.6091		
86.557	201	0.58458	0.218961	1.10919		
109.12	203	-	-	0.169913		



Figure 6 - Dependencies of conductivity change on potential difference (a) and resistance change on potential difference (b)

The shape of conducting surface is important because it affects the movement of electric charge in a material. The shape of particles that conduct electricity may be, for example, acicular, cylindrical, plate, tubular or spherical. These are idealized shapes of particles; actual shape can deviate more or less strongly from the ideal, depending on manufacturing process. Conducting surface in the form of array of identical cylinders was investigated in this paper. Volt-ampere characteristics were recorded from array of nanotubes with area of 1.5 cm². The scheme of the setup for measuring volt-ampere characteristics is following: a sample with deposited nanotubes is placed between two copper plates, which cover only the part of sample with nanotubes in film. Then plates are connected to current source with series connection of ammeter. Since PET-based template matrices are dielectric, they do not contribute to volt-ampere characteristics (VAC) throughout measuring range of voltage of matrix.

Figure 6 shows graphs of dependence of resistance changes and conductivity on potential difference applied during process of deposition.

The conductivity of Zn nanotubes was calculated by the formula (3):

$$\sigma = \frac{dI}{dU}\frac{l}{A},$$
(3)

where I – length of nanotubes, dI/dU – slope of I–U, A – area of conducting surface calculated using the formula (4):

$$A = (S_{out} - S_{inn}) * N * h,$$
 (4)

where S_{out} – area of outer ring of nanotubes (under section of nanotubes perpendicular to its axis), S_{inn} – area of inner ring, N – pore density (fluence of irradiation), h – height of nanotubes.

Figure 6 demonstrates that, according to EDA and XRD analysis, the crystal structure changes with increase of potential difference during deposition of Zn nanotubes and change in crystal structure leads to increasing conductivity. Change of

conductive properties of Zn nanotubes can be explained by heterogeneity of crystallites formed during electrochemical deposition. The number of defects that hinder the flow of electrons decreases with decrease of average size of crystallites in nanotubes. The number of defects directly affects conductive properties of Zn nanotubes.

4. Conclusion

This paper describes the method of growth of Zn-based nanotubes using electrochemical deposition. Track-etched PETbased membranes with pore density of 4.0E+07, thickness of 12 microns and pore diameters of 380 nm, were used as templates. Characterization of the structural features of obtained nanotubes was carried out by SEM, EDA and XRD. According to EDA analysis, formation of oxide phase of zinc with atomic ratio Zn₇₃O₂₇ occurs as a results of process of deposition at potential difference 1.25 V. Oxide phase is displaced with increase of applied potential difference, which is due to the predominance of half of redox potential of Zn reduction over the formation of oxide phase ZnO. Thus, pure zinc nanotubes without additives of oxide phases can be prepared by electrochemical deposition at potential difference of 1.75 V. Calculations of average sizes of crystallites show that decrease of intensity of ZnO oxide phase leads to the decrease of crystallites sizes from 41.52 nm for 1.25 V to 29.34 nm for 1.75 V. The number of defects that hinder the flow of electrons decreases with decrease of average size of crystallites in nanotubes. The number of defects directly affects conductive properties of Zn nanotubes. Thus, production of Zn nanotubes opens up an opportunity of their application as semiconductor materials in electronic and optoelectronic circuits.

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